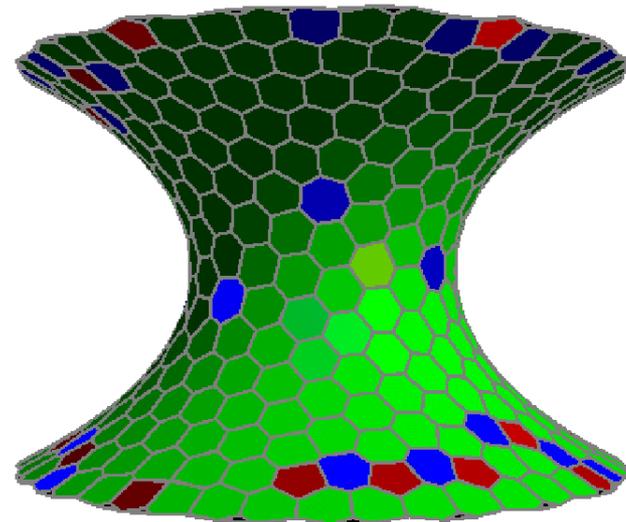
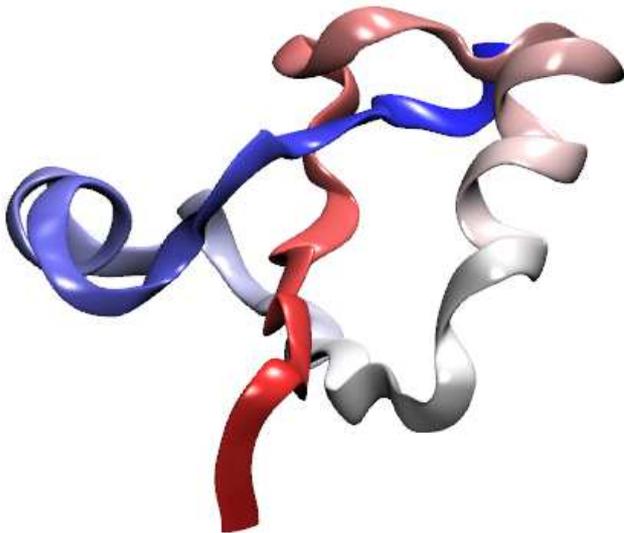


# Energy Landscapes: Molecules, Nanodevices and Machine Learning

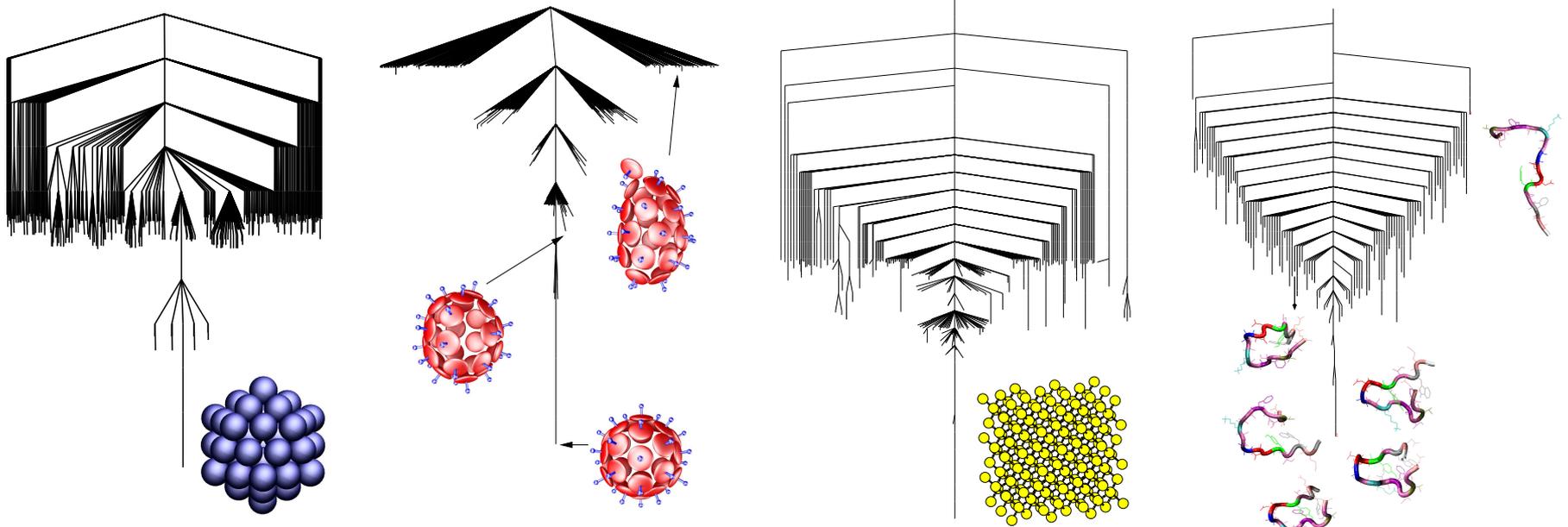
Objective: to exploit **stationary points** (minima and transition states) of the PES as a computational framework (*J. Phys. Chem. B*, **110**, 20765, 2006):

- **Basin-hopping** for global optimisation (*J. Phys. Chem. A*, **101**, 5111 1997)
- **Basin-sampling** for global thermodynamics (*J. Chem. Phys.*, **124**, 044102, 2006)
- **Discrete path sampling** for global kinetics (*Mol. Phys.*, **100**, 3285, 2002)

For small molecules all the relevant **stationary points** can be located. Larger systems, such as **proteins** (**left**) and **colloids** (**right**) require sampling.

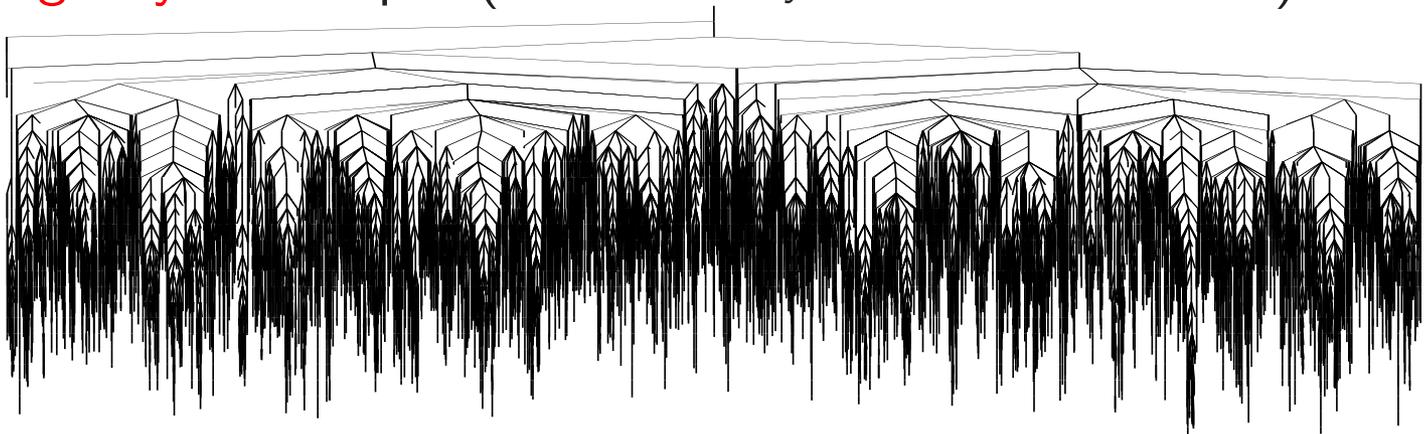


# Self-Organisation is Encoded in Single Funnel Landscapes

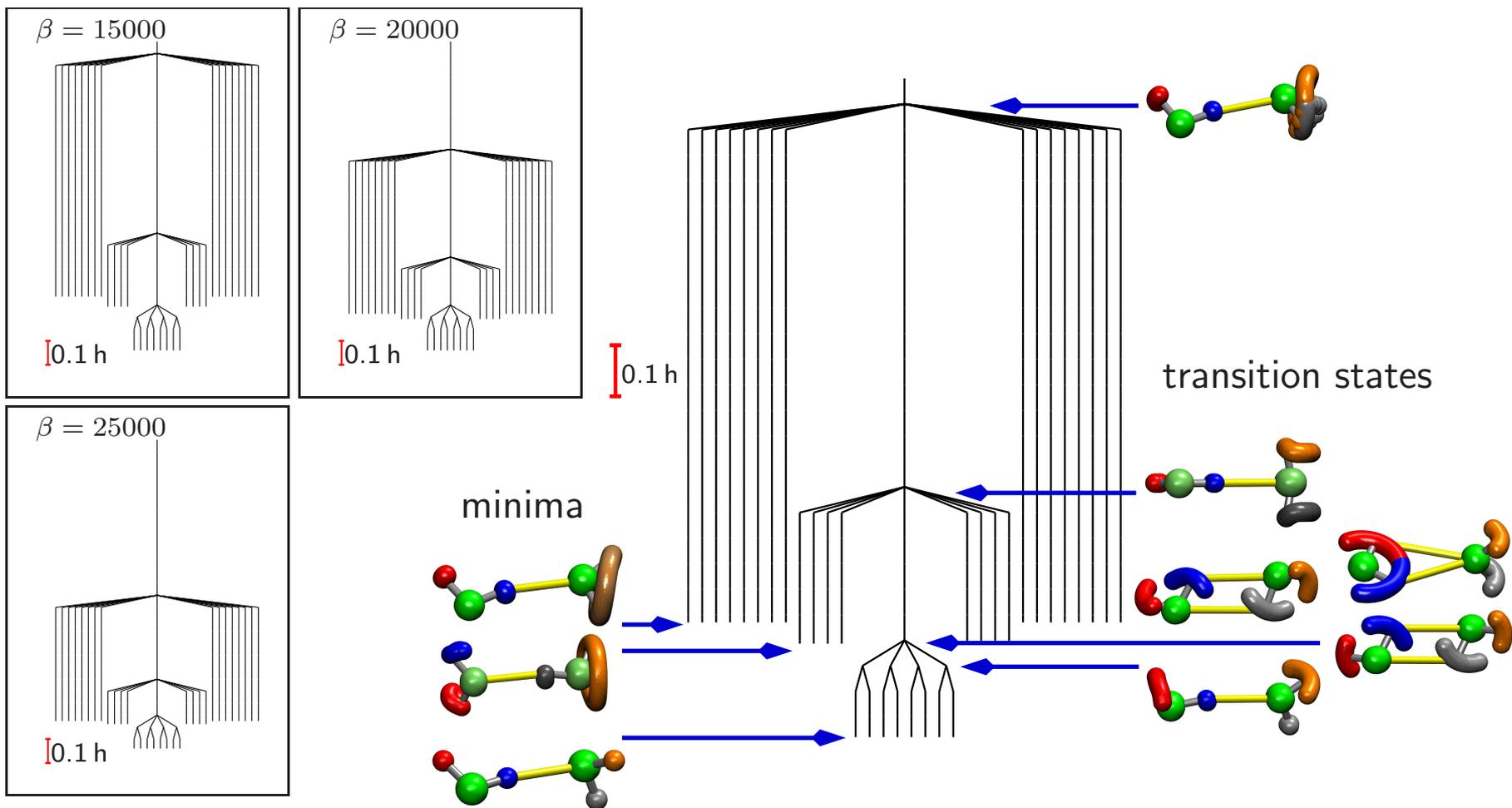


(Above) Energy landscapes for systems with **self-organising** properties. The **LJ<sub>55</sub> cluster**, an icosahedral **shell**, crystalline **silicon**, and the GB1 **peptide**.

(Below) A **glassy** landscape. (*Phil. Trans. Roy. Soc. A*, **363**, 357, 2005).

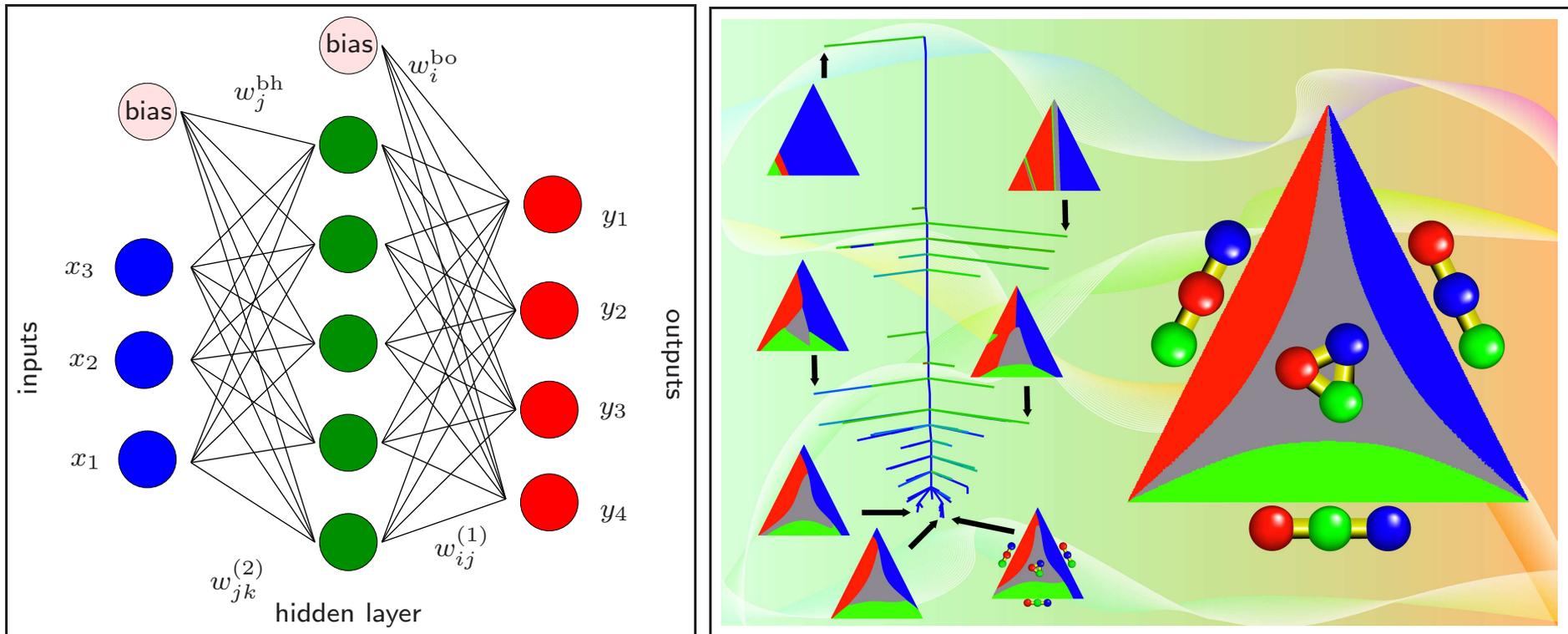


# Path Integral Energy Landscapes for Water Dimer



The ring polymer landscape of  $(\text{H}_2\text{O})_2$  includes classical and delocalised minima and transition states for the MBPOL potential (results for 201 beads).

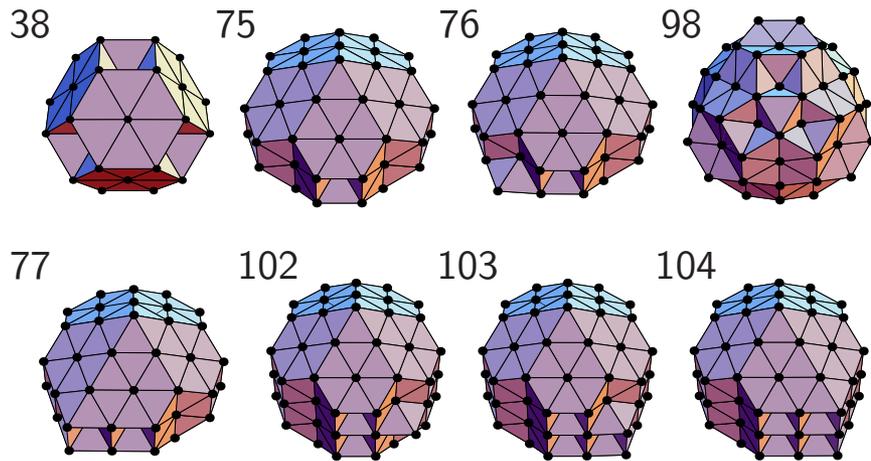
# Machine Learning Landscapes (*JCP*, **144**, 124119, 2016; *CPL*, **667**, 158, 2017)



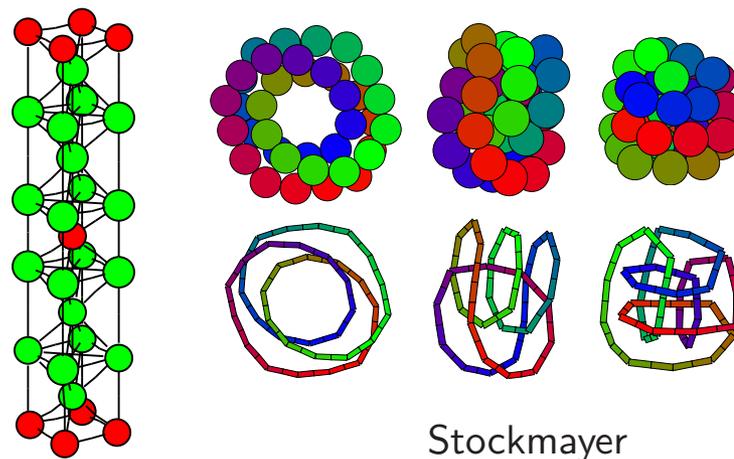
**Neural network** fits produce multiple solutions, defining a **landscape** for stationary points of the **cost function** for **multinomial logistic regression**.

In this example we predict the outcome of **geometry optimisation** for an **atomic cluster** with four distinct **isomers** using only the three initial **bond lengths**.

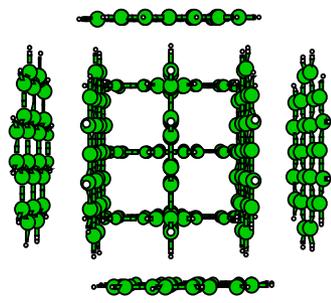
# Basin-Hopping Global Optimisation (*J. Phys. Chem. A*, 101, 5111, 1997)



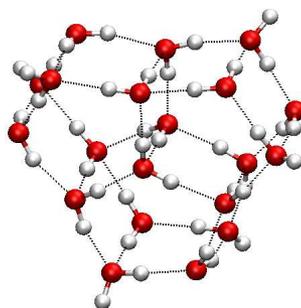
Non-icosahedral Lennard-Jones Clusters



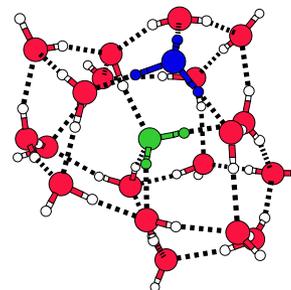
Binary LJ unit cell



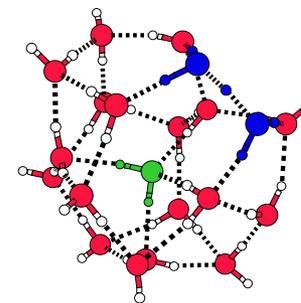
coronene<sub>10</sub>



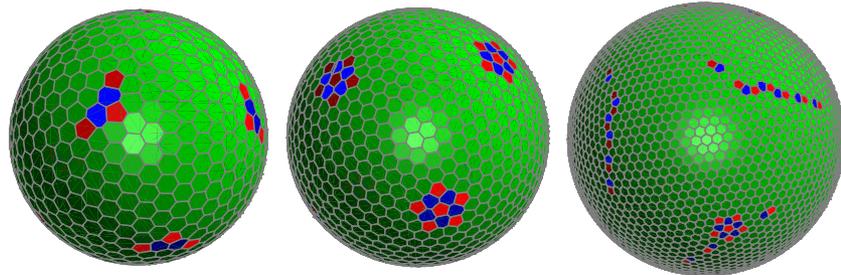
(H<sub>2</sub>O)<sub>20</sub>



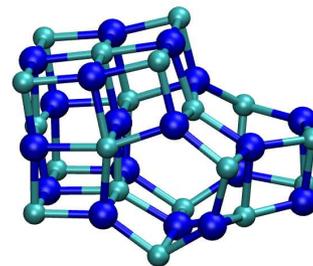
H<sub>3</sub>O<sup>+</sup>(H<sub>2</sub>O)<sub>20</sub> Eigen



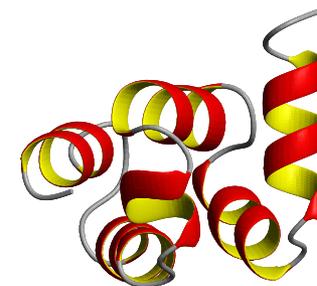
H<sub>3</sub>O<sup>+</sup>(H<sub>2</sub>O)<sub>20</sub> Zundel



Thomson problem



(NaCl)<sub>18</sub>Na<sup>+</sup>



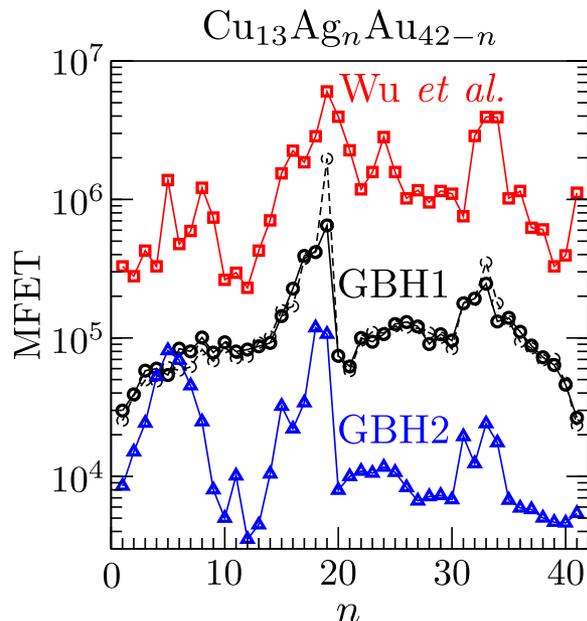
HYPA/FBP11

# Generalised Basin-Hopping (*PRL*, **113**, 156102, 2014; *PCCP*, **17**, 902, 2016)

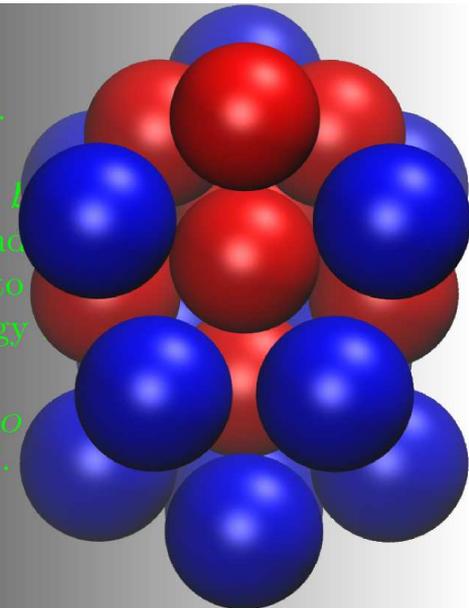
In **GBH** we focus on **biminima**, defined as structures where the energy cannot be lowered by **interchanging** inequivalent particles and **requenching**.

For **multicomponent** nanoalloy metal clusters fewer biminima exist for **segregated** systems with **lower** lattice mismatch.

In **GBH1** the **biminimum** property is **verified** by scanning the local neighbourhood, while in **GBH2** a subset is considered. (*PCCP*, **17**, 28331, 2015)



```
while  $C$  is FALSE:  
  Update  $C$  to TRUE.  
  Set  $\mathcal{A}' \leftarrow \mathcal{A}$  and  $\mathcal{B}' \leftarrow \mathcal{B}$ .  
  do  $\min(N_A, N_B)$  times:  
    Pick  $a'_I \in \mathcal{A}'$  and  $b'_J \in \mathcal{B}'$ .  
    Remove  $a'_I$  from  $\mathcal{A}'$  and  $b'_J$  from  $\mathcal{B}'$ .  
    Move  $a'_I$  to  $\mathcal{B}$  and  $b'_J$  to  $\mathcal{A}$ .  
    Update the total energy  $E$ .  
    if  $E < E_O$  then:  
      Update  $E_O \leftarrow E$ ,  $\mathcal{A}_O \leftarrow \mathcal{A}$ ,  $\mathcal{B}_O \leftarrow \mathcal{B}$ .  
      Reset  $C$  to FALSE.  
    end if.  
  end do loop.  
end while loop.
```



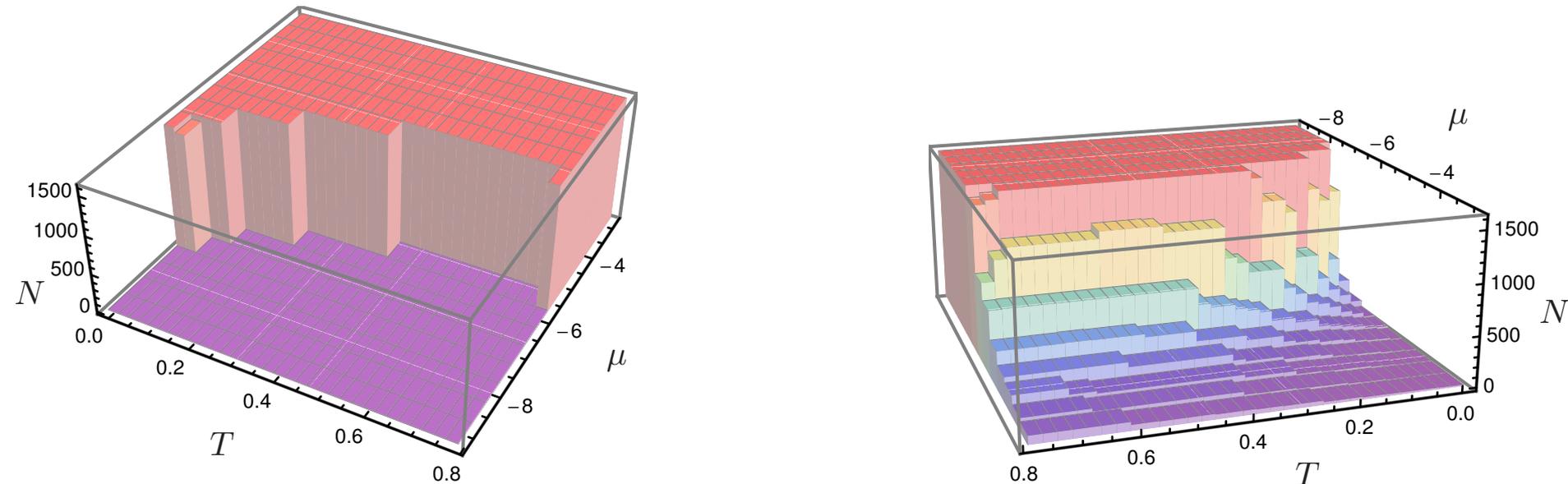
# Grand and Semi-Grand Canonical Basin-Hopping (*JCTC*, 12, 902, 2016)

The **accept/reject** condition for **GCBH** employs the local **grand potential**:

$$\xi_\alpha = V_\alpha - \mu N - k_B T \ln \frac{n_\alpha \sqrt{8\pi} |\mathbf{I}_\alpha|^{1/2} (k_B T)^{3/2}}{\hbar^3 (\beta \hbar \bar{v}_\alpha)^\kappa},$$

which includes the **rigid rotor** partition function, with inertia tensor  $\mathbf{I}_\alpha$ .

**Blocks** of conventional BH steps are employed between changes of  $N$ .

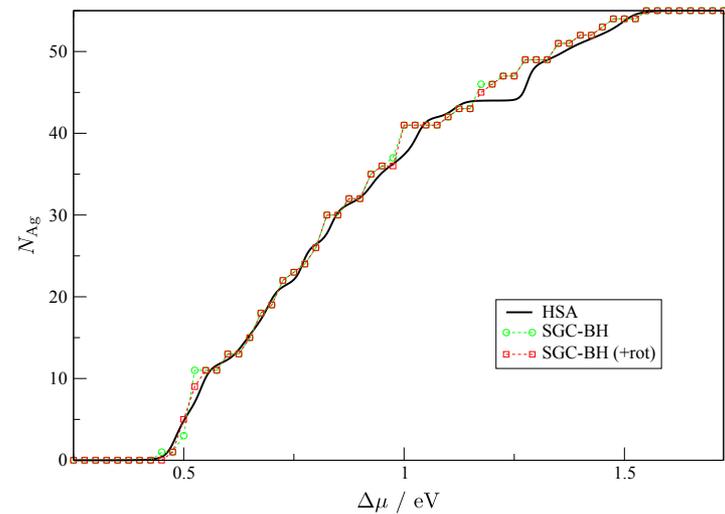
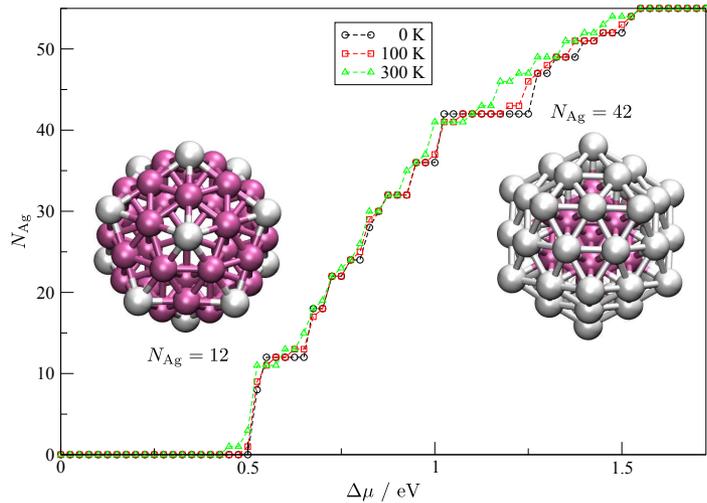


The **most** probable cluster size (left) corresponds to the **smallest** or **largest** size permitted. However, the **least** probable size can occur between the limits (right), suggesting a free energy **barrier** to **nucleation** and cluster growth.

For a **binary semigrand** canonical potential we consider **fixed**  $N = N_A + N_B$  and **variable**  $N_A$  and  $N_B$  with potential

$$\xi_\alpha = V_\alpha - \Delta\mu N_B - k_B T \ln \frac{(k_B T)^{3/2} n_\alpha \sqrt{8\pi} |\mathbf{I}_\alpha|^{1/2}}{\hbar^3 (\beta \hbar \bar{\nu}_\alpha)^\kappa}.$$

$\Delta\mu = \mu_B - \mu_A$  is the chemical potential **difference**;  $n_\alpha = 2N_A!N_B!/o_\alpha$ .



**Left:** the most favourable composition for icosahedral  $\text{Ag}_n\text{Pd}_{55-n}$  as a function of  $\Delta\mu$  exhibits **steps** at  $N_{\text{Ag}} = 12$  and  $N_{\text{Ag}} = 42$ .

**Right:** **lumping** probabilities for minima with the same composition and including non-icosahedral structures **smooths** the steps.

## Basin-Sampling for Global Thermodynamics (CPL, 584, 1, 2013)

Broken ergodicity is treated using basin-hopping, while the configuration space corresponding to high temperature is sampled by parallel tempering.

We define a two-dimensional density of states using systematic local minimisation, and couple these statistics to a model anharmonic form connected to the low-temperature limit for structures obtained via global optimisation.

The number of visits to quench potential energy bin  $q$  from instantaneous potential energy bin  $i$  in replica  $r$  is  $\mathcal{N}_{iqr}$ : a two-dimensional histogram.

The corresponding canonical probability distribution for replica  $r$  is

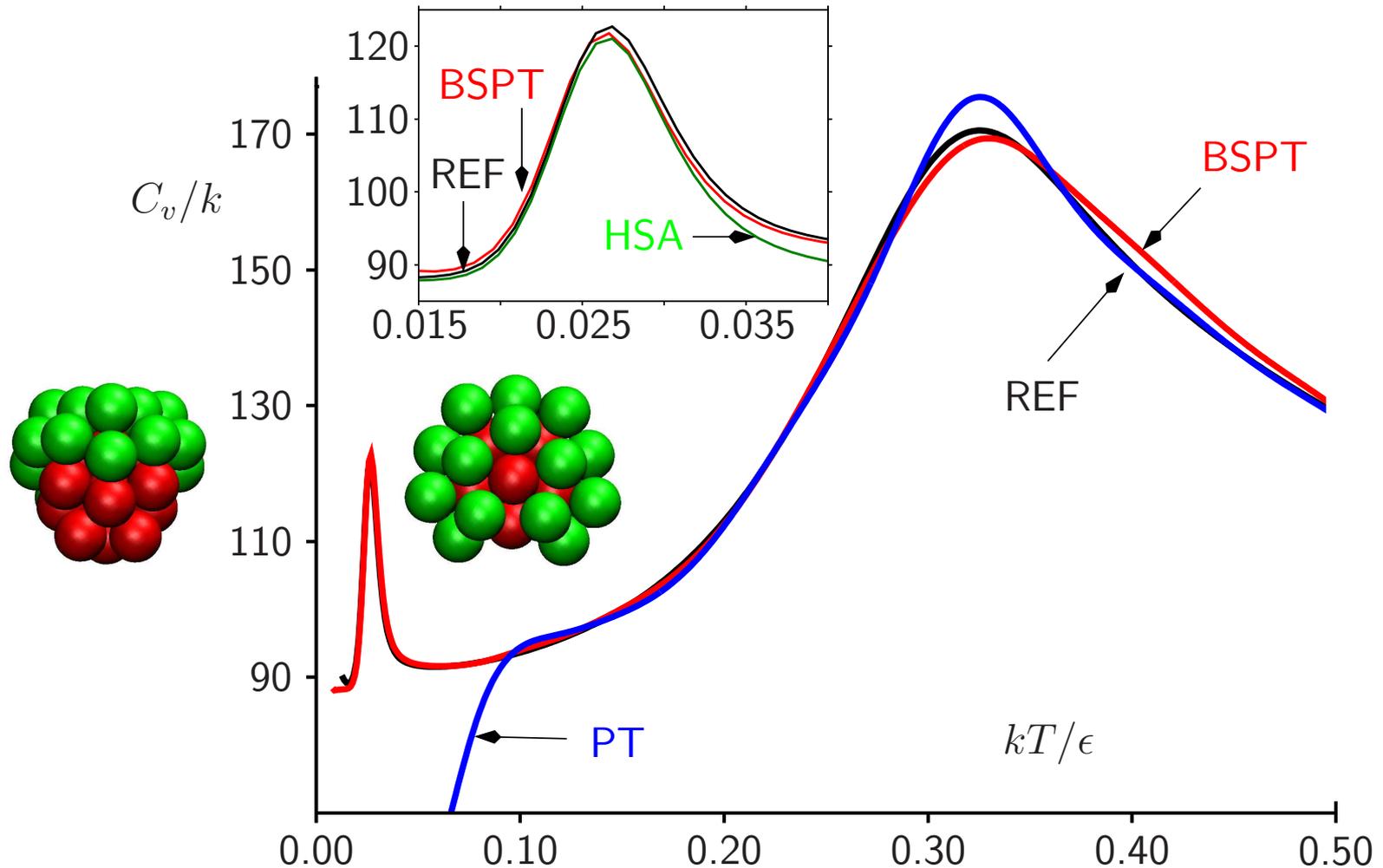
$$P(V_i^I, V_q^Q, T_r) = \mathcal{N}_{iqr} / \mathcal{N}_r \propto \Omega_c(V_i^I, V_q^Q) e^{-V_i^I / kT_r}.$$

The analytical density of states for a Morse potential suggests a model anharmonic representation for each quench bin with two fitting parameters:

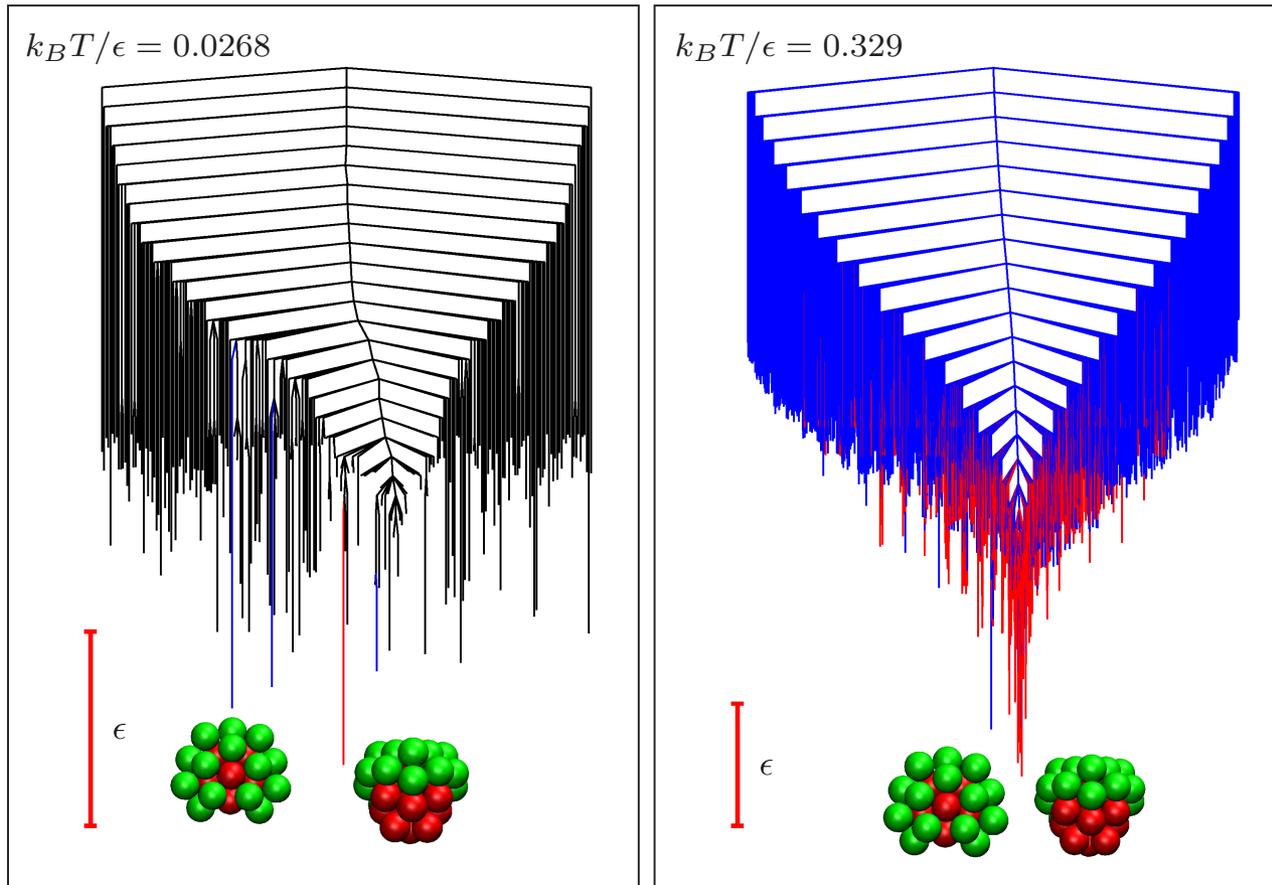
$$\ln \Omega_c(V_i^I, V_q^Q) = [(3N - 6)/2 - 1 + e^{A_q} (V_i^I - V_q^Q)] \ln (V_i^I - V_q^Q) + B_q.$$

Accurate thermodynamics were obtained for the **solid-solid** phase transition in **LJ<sub>31</sub>** with  $10^6$  equilibration steps (discarded),  $5 \times 10^6$  parallel tempering steps, and  $5 \times 10^6$  BSPT steps using 24 replicas and **quenching** every 30 steps.

This run took **21.8 minutes** compared to **110.5 hours** for **parallel tempering**.



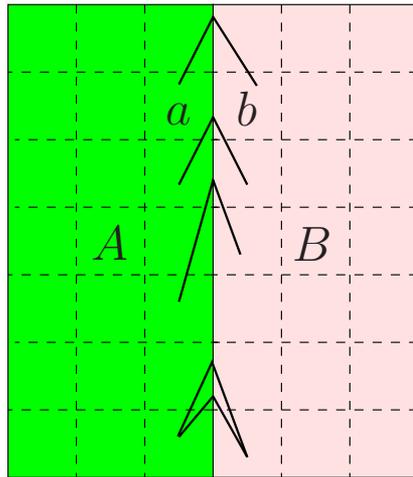
# Assigning Heat Capacity Features ( *Phys. Rev. E*, **95**, 030105R, 2017 )



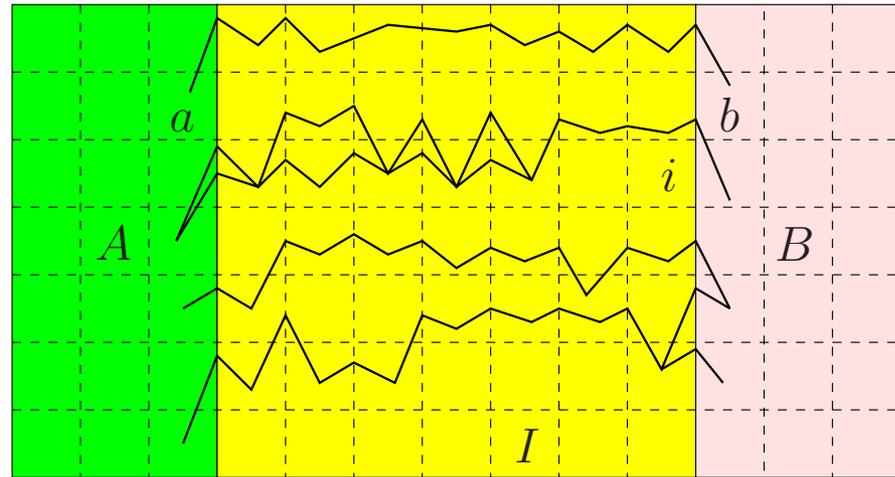
Contributions to  $C_V$  can be decomposed as sums over local minima with **positive** and **negative** occupation probability gradients,  $g_\gamma(T) = \partial p_\gamma(T) / \partial T$ :

$$C_V = \kappa k_B + k_B T^2 \sum_{\gamma}^{g_\gamma(T) > 0} g_\gamma(T)^2 / p_\gamma(T) + k_B T^2 \sum_{\gamma}^{g_\gamma(T) < 0} g_\gamma(T)^2 / p_\gamma(T)$$

# Discrete Path Sampling (*Mol. Phys.*, **100**, 3285, 2002; **102**, 891, 2004).



no intervening minima



$$\frac{p_a(t)}{p_{a'}(t)} = \frac{p_a^{\text{eq}}}{p_{a'}^{\text{eq}}} \quad \dot{p}_i(t) = 0 \quad \frac{p_b(t)}{p_{b'}(t)} = \frac{p_b^{\text{eq}}}{p_{b'}^{\text{eq}}}$$

Phenomenological  $A \leftrightarrow B$  rate constants can be formulated as sums over **discrete paths**, defined as sequences of local minima and the transition states that link them, weighted by equilibrium occupation probabilities,  $p_b^{\text{eq}}$ :

$$k_{AB}^{\text{SS}} = \frac{1}{p_B^{\text{eq}}} \sum_{a \leftarrow b} P_{ai_1} P_{i_1 i_2} \cdots P_{i_{n-1} i_n} P_{i_n b} \tau_b^{-1} p_b^{\text{eq}} = \frac{1}{p_B^{\text{eq}}} \sum_{b \in B} \frac{C_b^A p_b^{\text{eq}}}{\tau_b},$$

where  $P_{\alpha\beta}$  is a **branching probability** and  $C_b^A$  is the **committor** probability that the system will visit an  $A$  minimum **before** it returns to the  $B$  region.

## Rates from Graph Transformation (*JCP*, **124**, 234110, 2006; **130**, 204111, 2009)

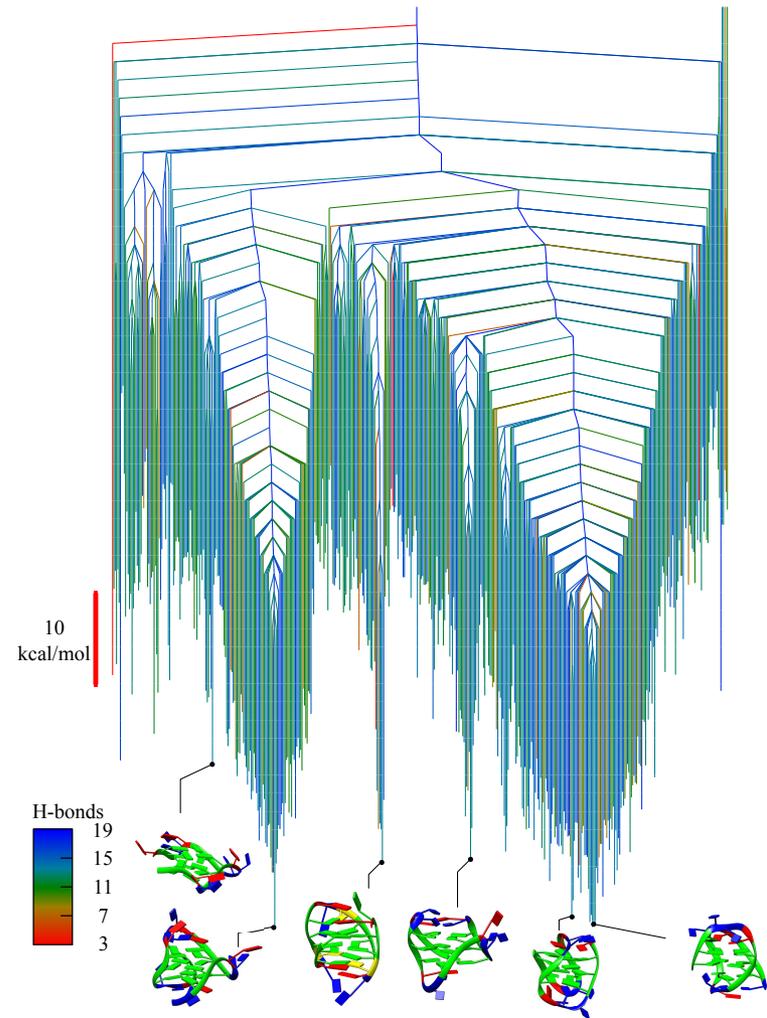
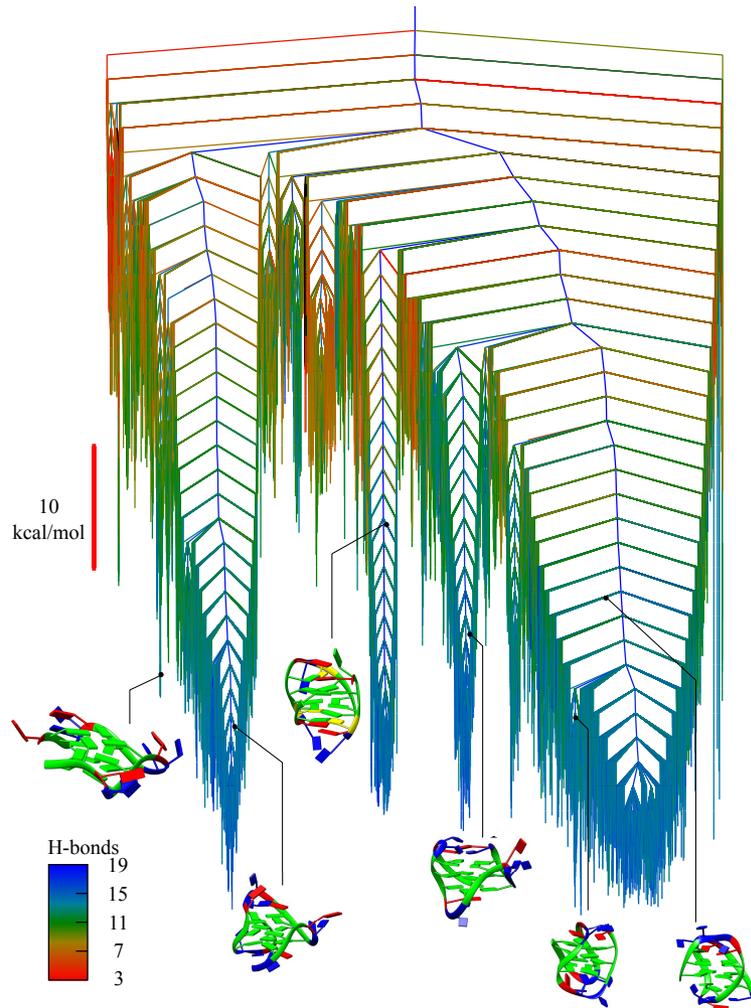
The deterministic **graph transformation** procedure is **non-stochastic** and **non-iterative**. Minima,  $x$ , are progressively **removed**, while the branching probabilities and waiting times in adjacent minima,  $\beta$ , are **renormalised**:

$$P'_{\gamma\beta} = P_{\gamma\beta} + P_{\gamma x} P_{x\beta} \sum_{m=0}^{\infty} P_{xx}^m = P_{\gamma\beta} + \frac{P_{\gamma x} P_{x\beta}}{1 - P_{xx}}, \quad \tau'_{\beta} = \tau_{\beta} + \frac{P_{x\beta} \tau_x}{1 - P_{xx}}.$$

Each transformation **conserves** the **MFPT** from every reactant state to the set of product states with an execution time **independent** of temperature:

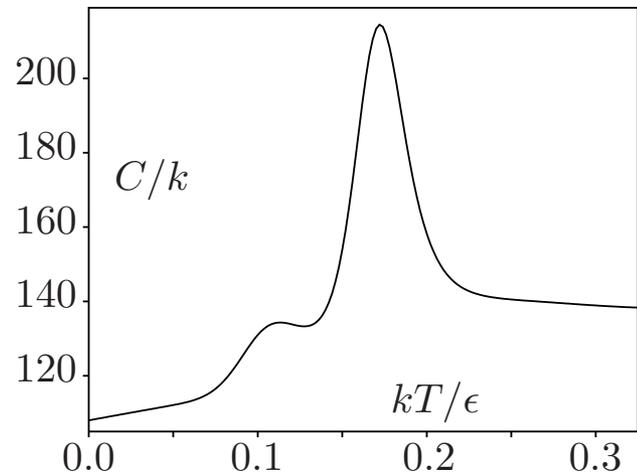
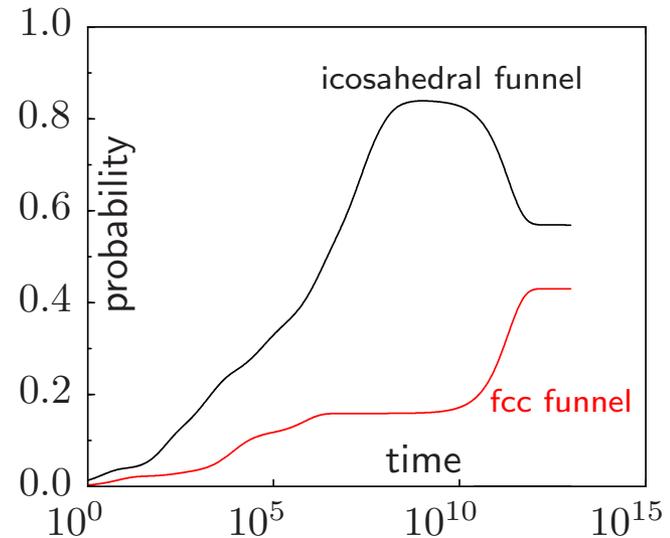
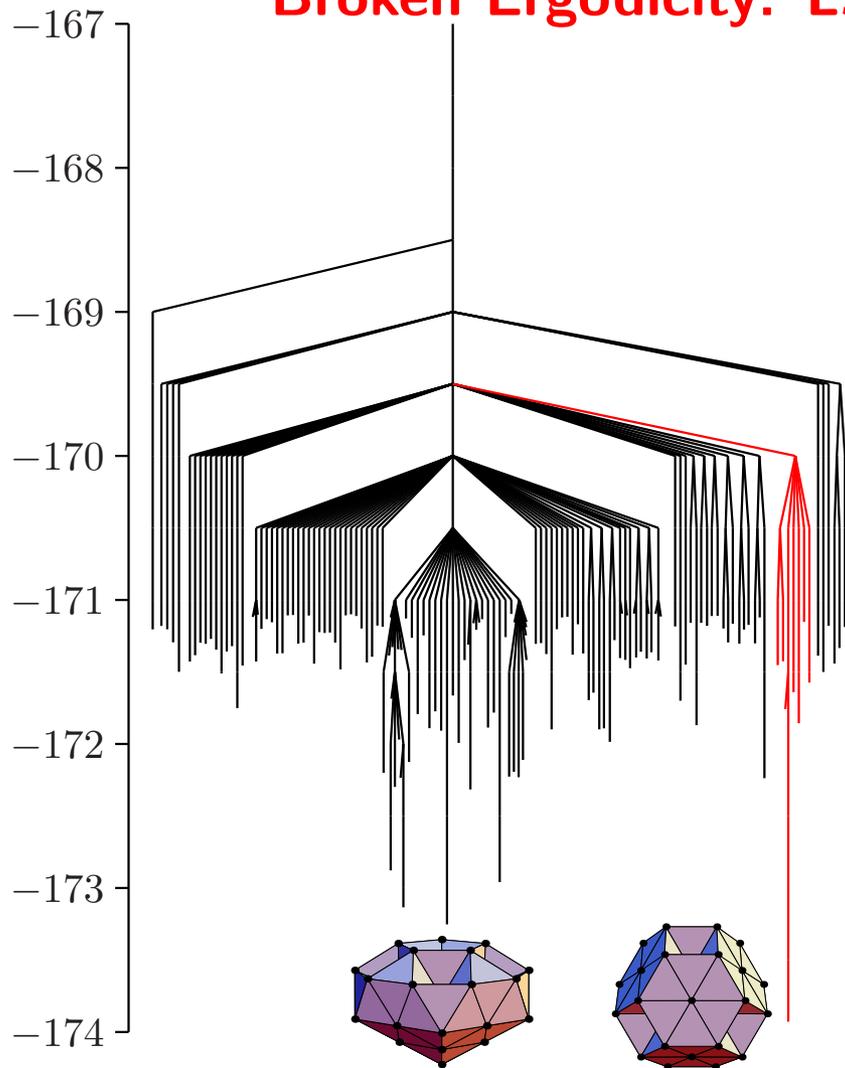
$kT/K$	$\Delta F_{\text{barrier}}$	$N_{\text{min}}$	$N_{\text{ts}}$	NGT/s	SOR/s	KMC/s
298	5.0	272	287	8	13	85,138
298	4.5	2,344	2,462	8	217,830	
1007	-	40,000	58,410	35	281	1,020,540
1690	-	40,000	58,410	39	122,242	

# Landscapes for a DNA four-fold telomere repeat: $A(G_3TTA)_3G_3$



**G-quadruplexes**, stabilised by **quartets** of **guanine** bases, decrease the activity of **telomerase**, which maintains the **length** of telomeric repeats. **Misfunctioning** is associated with numerous **cancer** conditions. (*JCP*, **147**, 152715, 2017)

## Broken Ergodicity: $\text{LJ}_{38}$ (*Phys. Rev. E*, **60**, 3701, 1999)



$\text{LJ}_{38}$  exhibits a **double funnel** due to competition between icosahedral and truncated **octahedral** morphologies. The interconversion rate for  $\text{Ar}_{38}$  is calculated as  $55 \text{ s}^{-1}$  at 14 K where a **solid-solid** transition occurs.

## Simulating structural transitions by direct transition current sampling: The example of LJ<sub>38</sub>

Massimiliano Picciani,<sup>1,a)</sup> Manuel Athènes,<sup>1</sup> Jorge Kurchan,<sup>2</sup> and Julien Tailleur<sup>3</sup>

<sup>1</sup>*CEA, DEN, Service de Recherches de Métallurgie Physique, F-91191 Gif-sur-Yvette, France*

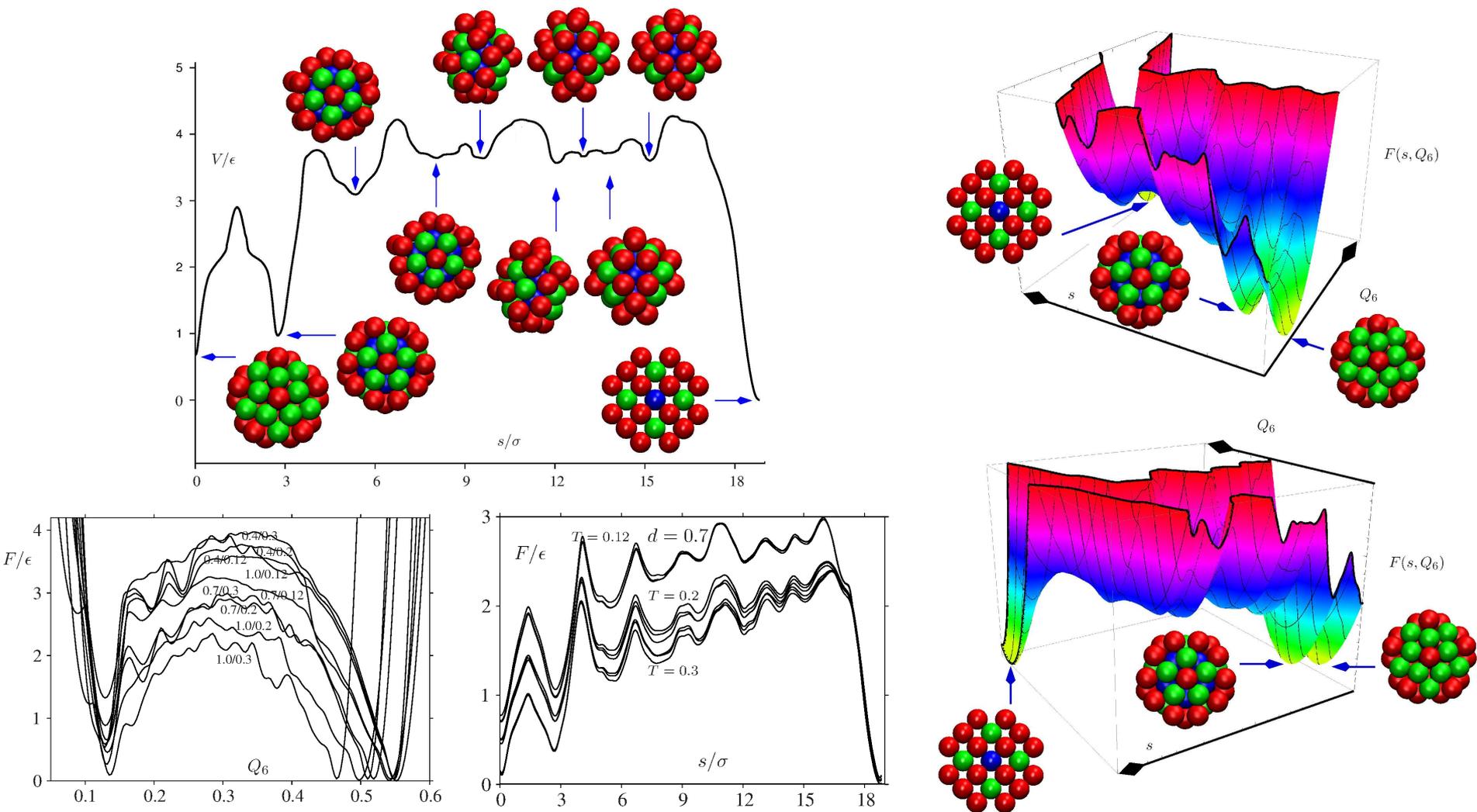
<sup>2</sup>*CNRS; ESPCI, 10 rue Vauquelin, UMR 7636 PMMH, 75005 Paris, France*

<sup>3</sup>*School of Physics of Astronomy, SUPA, University of Edinburgh, The King's Buildings, Mayfield Road, EH9 3JZ Edinburgh, United Kingdom*

(Received 2 March 2011; accepted 21 June 2011; published online 20 July 2011)

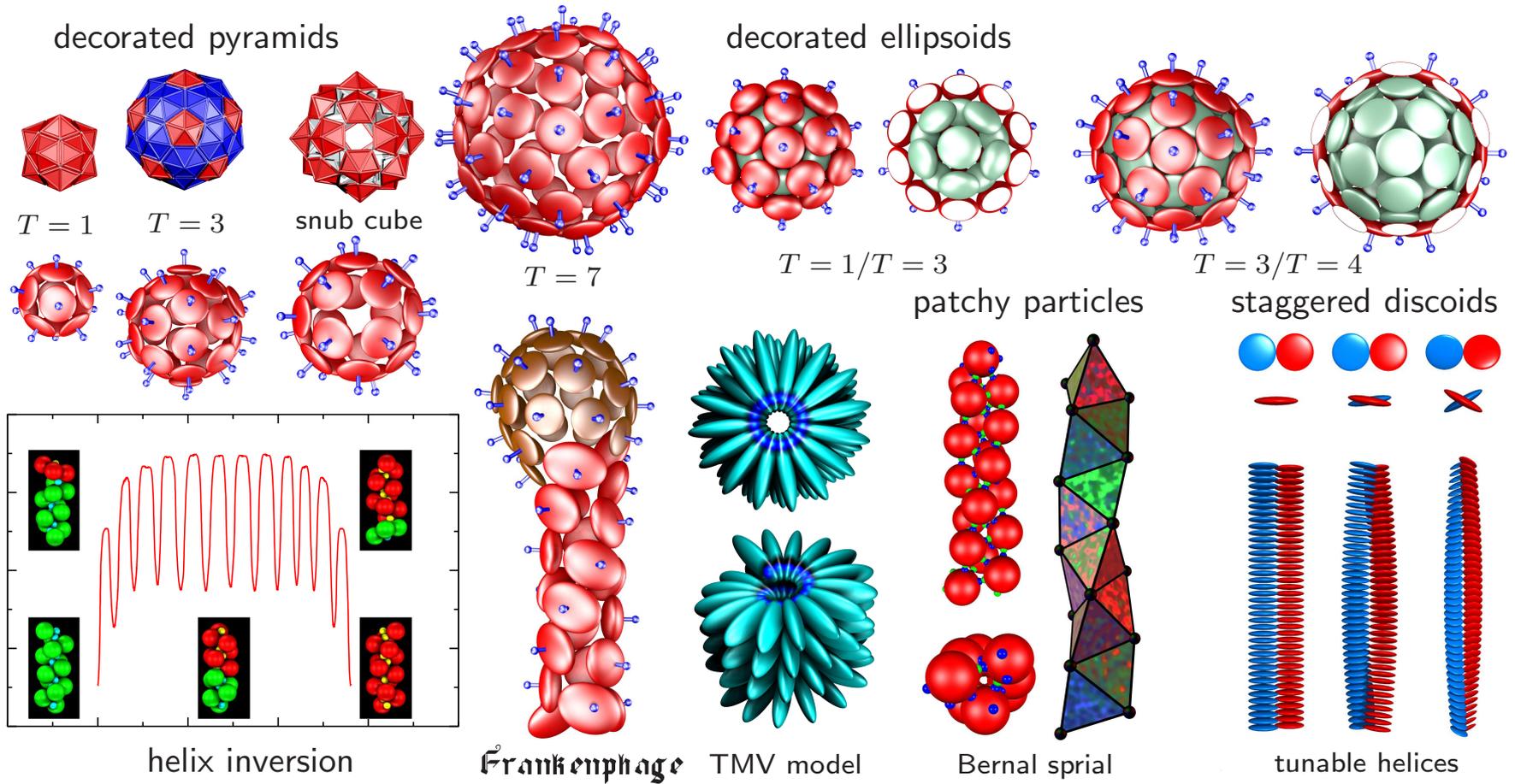
Another attempt to study the transitions between the two funnels of LJ<sub>38</sub> relies on the use of transition path sampling.<sup>33</sup> Because of the number of metastable states separating the two main basins, the traditional shooting and shifting algorithm failed here, despite previous success for smaller LJ clusters.<sup>39</sup> The authors thus developed a two-ended approach which manages to successfully locate reaction paths between the two basins: they started from a straight trial trajectory linking the two minima, and obtained convergence towards trajectories of energies similar to those obtained in the discrete path sampling approach.<sup>33</sup> Although the authors point out the lack of ergodicity in the sampling within their approach and the sensitivity on the “discretization” of the trajectories, this is nevertheless a progress and the main drawback remains the high computational cost (the work needed  $10^5$  h of central processing unit (cpu) time) to obtain such converged trajectories. In contrast, the simulations we present below required less than  $10^2$  h of cpu time.

# Free Energy Profiles for an LJ<sub>38</sub> Pathway (JCP, 142, 130901, 2015.)



Projection onto the **bond order** parameter  $Q_6$  averages over **surface reorganization** and other mechanistic details. However, the profile based on pathways defined **geometrically** faithfully reflects the underlying **barriers**.

# Coarse-Grained Models (*PCCP*, 11, 1970, 2009; *ACS Nano*, 4, 219, 2010)



The **angle-axis** formulation provides a particularly convenient framework for **mesoscopic** modelling, using both **decorated** rigid bodies and **ellipsoids**.

All the terms involving **angle-axis** coordinates can be obtained by the action of a **rotation matrix** and its derivatives, which are **system-independent**.

1st derivatives:  $\mathbf{R}_k \equiv \frac{\partial \mathbf{R}}{\partial p_k} = \frac{p_k \sin \theta}{\theta} \tilde{\mathbf{p}}^2 + (1 - \cos \theta)(\tilde{\mathbf{p}}_k \tilde{\mathbf{p}} + \tilde{\mathbf{p}} \tilde{\mathbf{p}}_k) + \frac{p_k \cos \theta}{\theta} \tilde{\mathbf{p}} + \sin \theta \tilde{\mathbf{p}}_k$ , with  $\tilde{\mathbf{p}}_1 = \frac{1}{\theta^3} \begin{pmatrix} 0 & p_1 p_3 & -p_1 p_2 \\ -p_1 p_3 & 0 & p_1^2 - \theta^2 \\ p_1 p_2 & \theta^2 - p_1^2 & 0 \end{pmatrix}$

2nd derivatives:  $\mathbf{R}_{kk} \equiv \frac{\partial^2 \mathbf{R}}{\partial p_k^2} = \frac{2p_k \sin \theta}{\theta} (\tilde{\mathbf{p}}_k \tilde{\mathbf{p}} + \tilde{\mathbf{p}} \tilde{\mathbf{p}}_k) + \left( \frac{p_k^2 \cos \theta}{\theta^2} - \frac{p_k^2 \sin \theta}{\theta^3} + \frac{\sin \theta}{\theta} \right) \tilde{\mathbf{p}}^2$   
 $+ (1 - \cos \theta)(2\tilde{\mathbf{p}}_k^2 + \tilde{\mathbf{p}}_{kk} \tilde{\mathbf{p}} + \tilde{\mathbf{p}} \tilde{\mathbf{p}}_{kk}) + \left( -\frac{p_k^2 \sin \theta}{\theta^2} - \frac{p_k^2 \cos \theta}{\theta^3} + \frac{\cos \theta}{\theta} \right) \tilde{\mathbf{p}} + \frac{2p_k \cos \theta}{\theta} \tilde{\mathbf{p}}_k + \sin \theta \tilde{\mathbf{p}}_{kk}$ ,

and  $\mathbf{R}_{kl} \equiv \frac{\partial^2 \mathbf{R}}{\partial p_k \partial p_l} = \frac{p_k \sin \theta}{\theta} (\tilde{\mathbf{p}}_l \tilde{\mathbf{p}} + \tilde{\mathbf{p}} \tilde{\mathbf{p}}_l) + \left( \frac{p_k p_l \cos \theta}{\theta^2} - \frac{p_k p_l \sin \theta}{\theta^3} \right) \tilde{\mathbf{p}}^2 + \frac{p_l \sin \theta}{\theta} (\tilde{\mathbf{p}}_k \tilde{\mathbf{p}} + \tilde{\mathbf{p}} \tilde{\mathbf{p}}_k)$   
 $+ (1 - \cos \theta)(\tilde{\mathbf{p}}_{kl} \tilde{\mathbf{p}} + \tilde{\mathbf{p}}_k \tilde{\mathbf{p}}_l + \tilde{\mathbf{p}}_l \tilde{\mathbf{p}}_k + \tilde{\mathbf{p}} \tilde{\mathbf{p}}_{kl}) - \left( \frac{p_k p_l \sin \theta}{\theta^2} + \frac{p_k p_l \cos \theta}{\theta^3} \right) \tilde{\mathbf{p}} + \frac{p_k \cos \theta}{\theta} \tilde{\mathbf{p}}_l + \frac{p_l \cos \theta}{\theta} \tilde{\mathbf{p}}_k + \sin \theta \tilde{\mathbf{p}}_{kl}$ .

Denote positions in the **body-fixed** frame by superscript 0. For rigid bodies  $I$  and  $J$  with sites  $i$  and  $j$  defining site-site **isotropic** potentials  $U_{ij}^{IJ}$  the **potential energy** is

$$U = \sum_I \sum_{J < I} \sum_{i \in I} \sum_{j \in J} f_{ij}(r_{ij}), \quad \text{where} \quad r_{ij} = |\mathbf{r}_{ij}| = |\mathbf{r}_i - \mathbf{r}_j| \quad \text{and} \quad f_{ij} \equiv U_{ij}^{IJ} \quad \text{so that}$$

$$\frac{\partial U}{\partial \zeta} = \sum_{J \neq I} \sum_{i \in I} \sum_{j \in J} f'_{ij}(r_{ij}) \frac{\partial r_{ij}}{\partial \zeta}, \quad \text{where} \quad f'_{ij} = \frac{df_{ij}(r_{ij})}{dr_{ij}}, \quad \frac{\partial r_{ij}}{\partial \mathbf{r}^I} = \hat{\mathbf{r}}_{ij}, \quad \frac{\partial r_{ij}}{\partial p_k^I} = \hat{\mathbf{r}}_{ij} \cdot \frac{\partial \mathbf{r}_{ij}}{\partial p_k^I} = \hat{\mathbf{r}}_{ij} \cdot (\mathbf{R}_k^I \mathbf{r}_i^0), \quad \mathbf{r}_{ij} = \mathbf{r}^I + \mathbf{R}^I \mathbf{r}_i^0 - \mathbf{r}^J - \mathbf{R}^J \mathbf{r}_j^0.$$

$$\frac{\partial^2 U_{ij}^{IJ}}{\partial r_k^I \partial r_l^J} = f_2(r_{ij}) r_{ij,k} r_{ij,l} \epsilon_{IJ} + f_1(r_{ij}) \delta_{kl} \epsilon_{IJ},$$

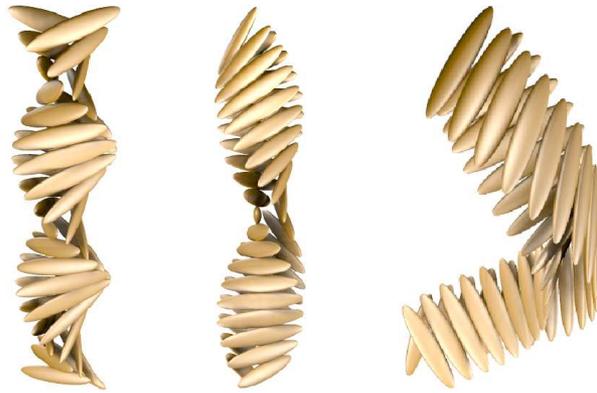
$$\frac{\partial^2 U_{ij}^{IJ}}{\partial p_k^I \partial p_l^J} = f_2(r_{ij}) (\mathbf{r}_{ij} \cdot \mathbf{R}_k^I \mathbf{r}_i^0) (\mathbf{r}_{ij} \cdot \mathbf{R}_l^J \mathbf{r}_j^0) \delta_{IJ} - f_2(r_{ij}) (\mathbf{r}_{ij} \cdot \mathbf{R}_k^I \mathbf{r}_i^0) (\mathbf{r}_{ij} \cdot \mathbf{R}_l^J \mathbf{r}_j^0) (1 - \delta_{IJ}) + f_1(r_{ij}) (\mathbf{R}_k^I \mathbf{r}_i^0) \cdot (\mathbf{R}_l^J \mathbf{r}_j^0) \delta_{IJ}$$

$$- f_1(r_{ij}) (\mathbf{R}_k^I \mathbf{r}_i^0) \cdot (\mathbf{R}_l^J \mathbf{r}_j^0) (1 - \delta_{IJ}) + f_1(r_{ij}) (\mathbf{r}_{ij} \cdot \mathbf{R}_{kl}^I \mathbf{r}_i^0) \delta_{IJ},$$

$$\frac{\partial^2 U_{ij}^{IJ}}{\partial r_k^I \partial p_l^J} = f_2(r_{ij}) (\mathbf{r}_{ij} \cdot \mathbf{R}_l^J \mathbf{r}_j^0) r_{ij,k} \delta_{IJ} - f_2(r_{ij}) (\mathbf{r}_{ij} \cdot \mathbf{R}_l^J \mathbf{r}_j^0) r_{ij,k} (1 - \delta_{IJ}) + f_1(r_{ij}) [\mathbf{R}_k^I \mathbf{r}_i^0]_l \delta_{IJ} - f_1(r_{ij}) [\mathbf{R}_l^J \mathbf{r}_j^0]_l (1 - \delta_{IJ}).$$

where  $f_1(r_{ij}) = f'_{ij}(r_{ij})/r_{ij}$ ,  $f_2(r_{ij}) = f''_{ij}(r_{ij})/r_{ij}$ ,  $\epsilon_{IJ} = 1$  for  $I = J$  and  $\epsilon_{IJ} = -1$  for  $I \neq J$ , and  $\delta_{IJ}$  is the Kronecker delta.

# Helical Bilayers From Frustrated Building Blocks (*JPCB*, 117, 7918, 2013)



bilayer filaments



magnetic bilayers

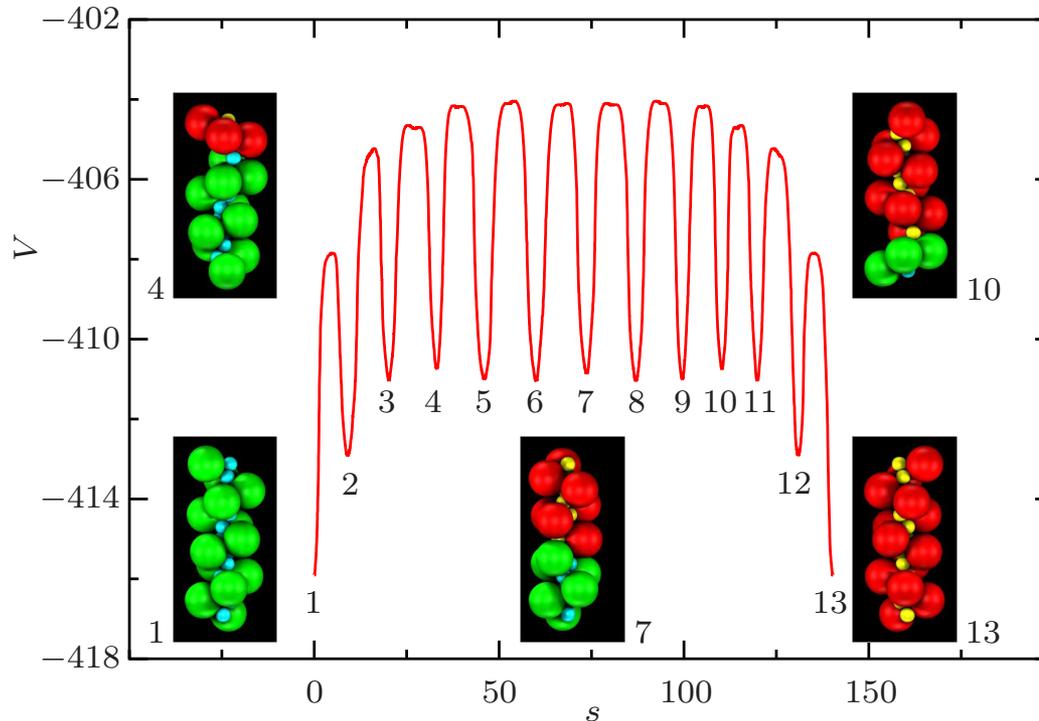
helical fibre morphologies

**Left:** introduction of a **cytochrome** domain into an **amyloid fibre** can change the morphology from **twisted** to **spiral** ribbons and induce systematic **kinking**.

**Centre:** rigid building blocks consisting of **two ellipsoids** can reproduce these structures, which are also observed for **Bauhinia seedpods**.

**Right:** the structure depends mostly on the **internal geometry** of the building blocks, rather than details of the **potential**. The **design** principles extend to **macroscopic** helices formed from elliptical **magnets**.

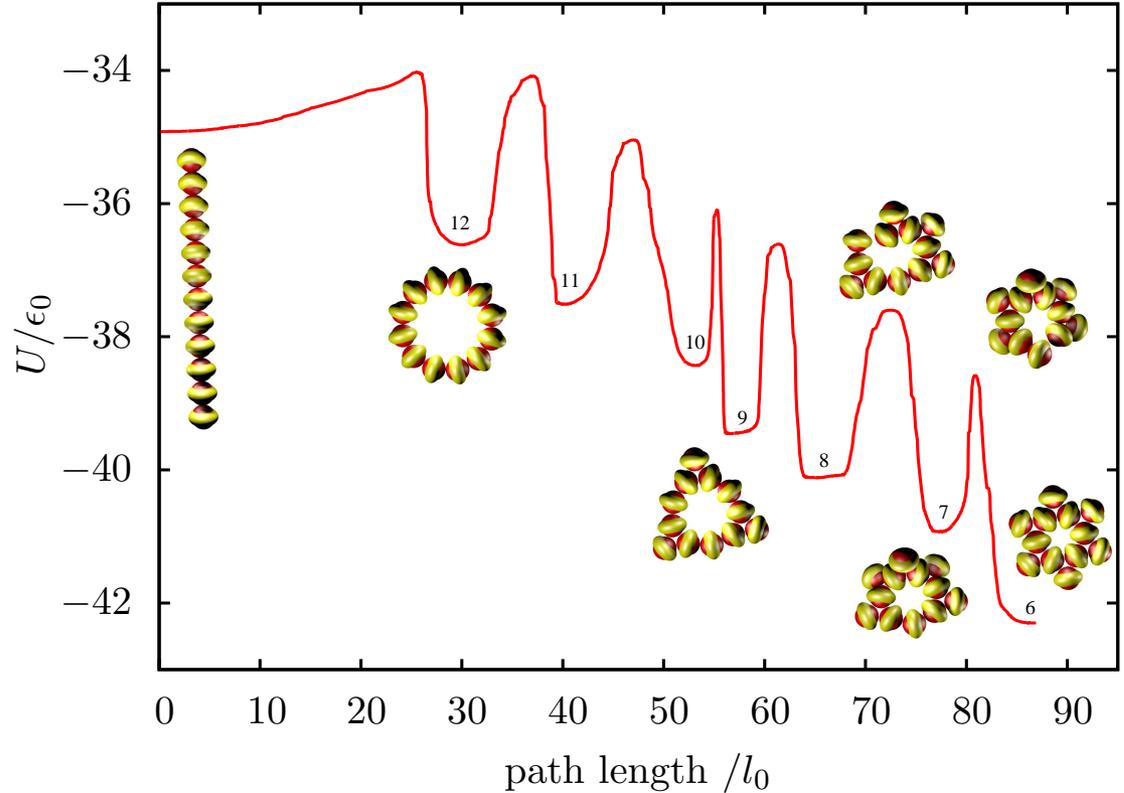
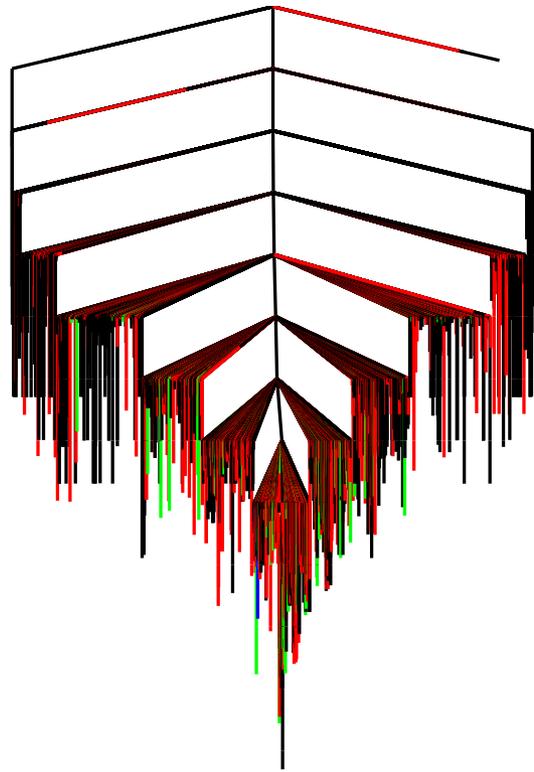
## A Nanodevice (*Soft Matter*, 7, 2325, 2011)



Coupled **linear** and **rotary** motion has been characterised for a helix composed of 13 asymmetric **dipolar dumbbells** in the presence of an **electric field**.

The helix changes **handedness** as the boundary between segments propagates along the strand via successive steps that switch the dumbbells.

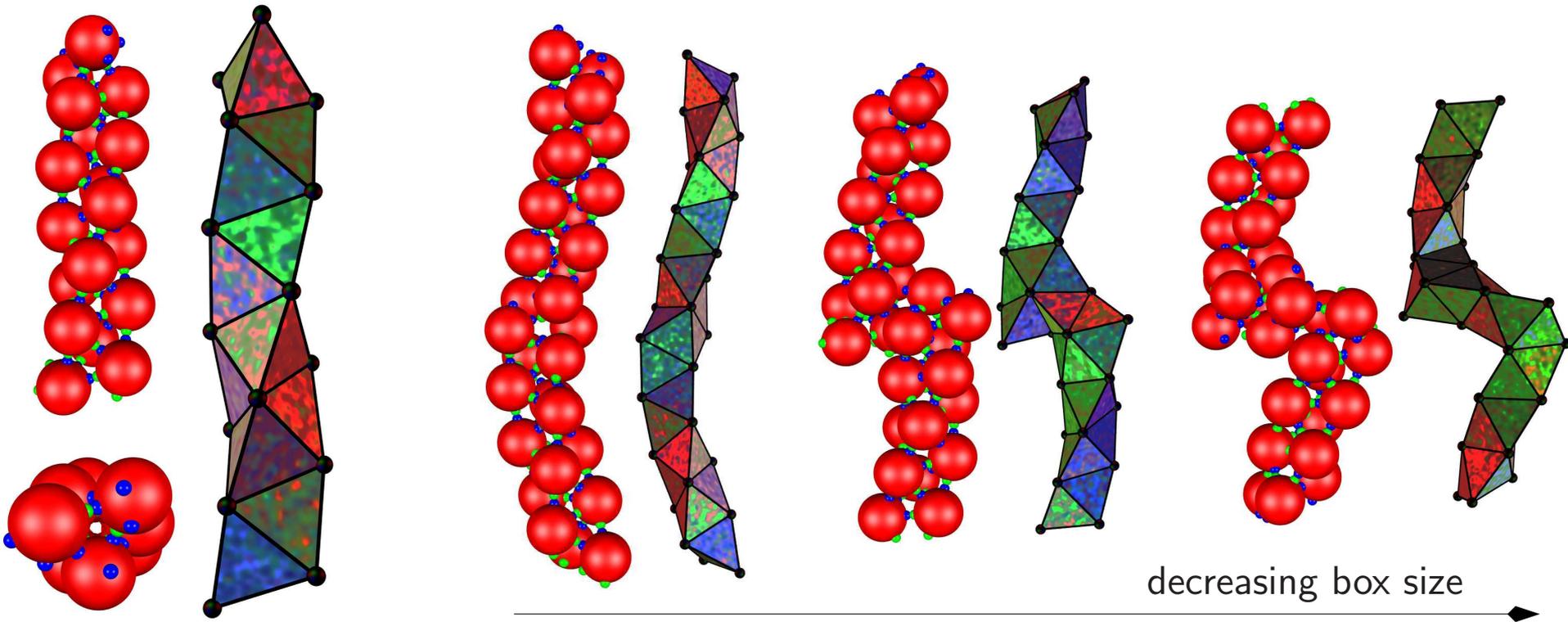
# Kagome Structures (Soft Matter, 11, 6663, 2015)



Energetically stabilised **Kagome** structures were designed using soft anisotropic triblock **Janus** particles.  This **unconstrained** model predicts that **sedimentation** effects **enhance** the stability.

**Rearrangements** between competing structures are highly **cooperative**.

## Designing a Bernal Spiral (ACS Nano, 7, 1246, 2013)



The simplest building blocks that support a Bernal spiral as the global minimum involve a **single** patch-antipatch pair **offset** by about  $10^\circ$  from linearity.

Left: Alternative views of a chiral Bernal spiral consisting of 18 particles.

Right: **compressed** spirals (30 particles, periodic boundaries) exhibit **supercoiling** or **breaks**, which resemble structures seen in **confocal microscopy**.

## Connecting Dynamics and Thermodynamics (*Science*, 293, 2067, 2001)

The organisation of a PES is governed by its **stationary points**, where Taylor expansions provide local descriptions in terms of **Hessian matrices**.

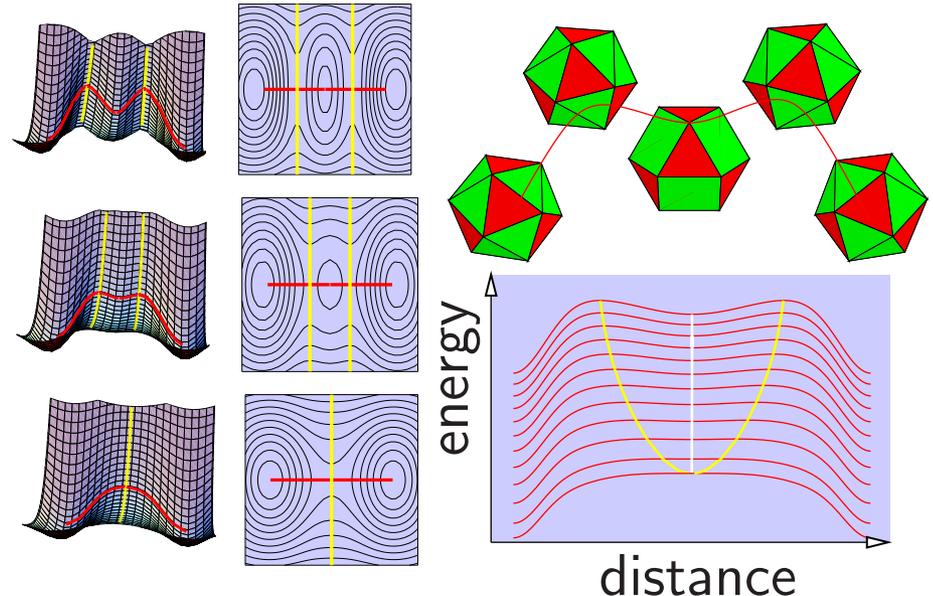
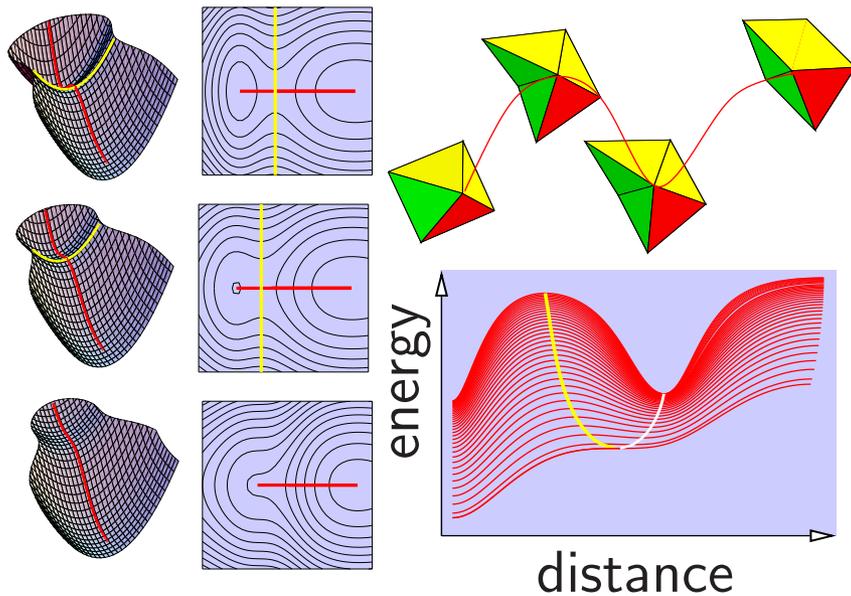
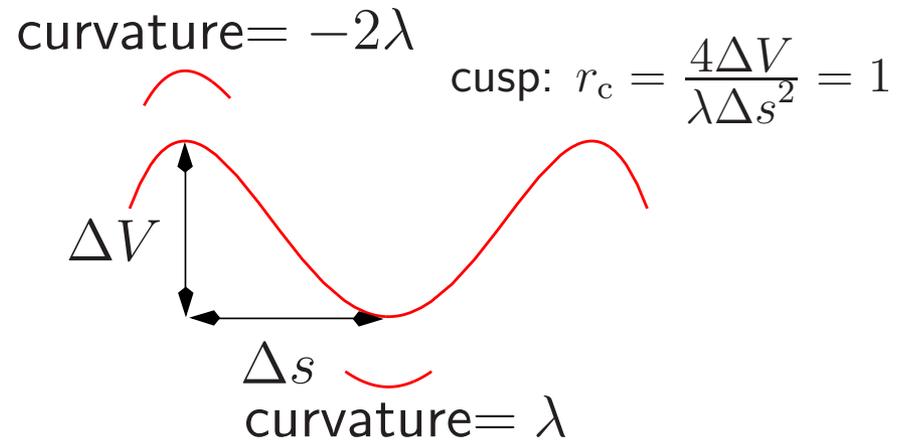
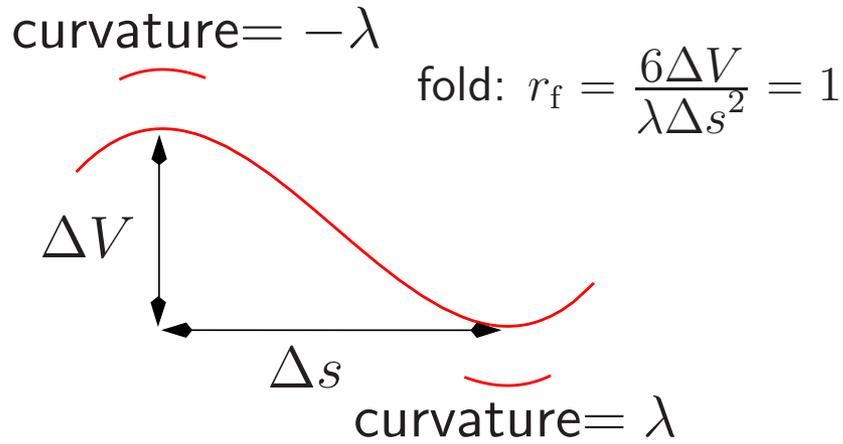
The organisation of **families** of PES's as a function of **parameters** in the potential is determined by the stationary points that possess additional zero Hessian eigenvalues, known as **non-Morse** points.

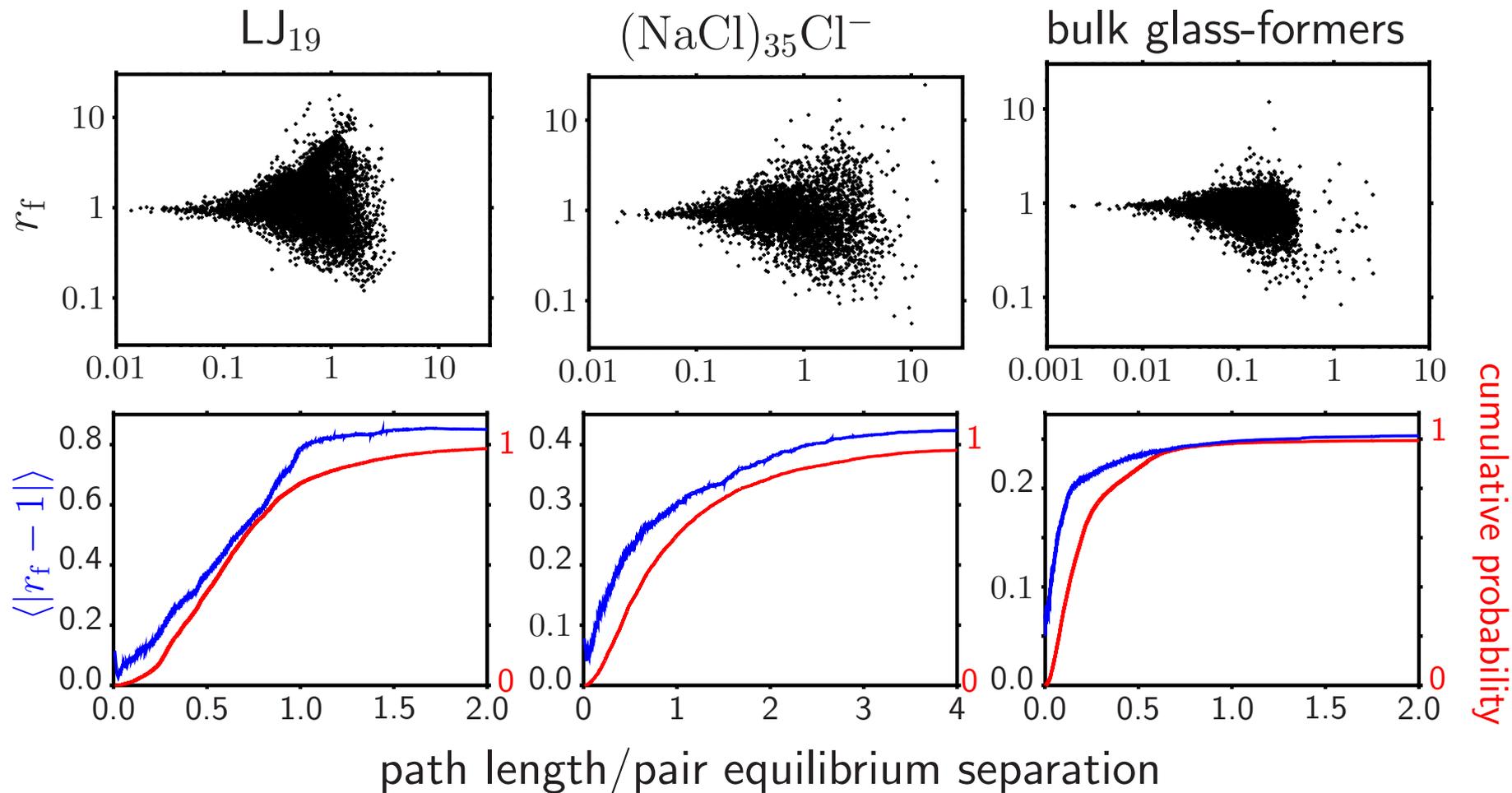
**Catastrophe theory** provides a local representation of the PES around non-Morse points as a function of **both** atomic coordinates and parameters.

The **splitting lemma** reduces the dimensionality to the **essential** variables, while **transversality** guarantees that the resulting classifications are **universal**.

The simplest one-parameter catastrophes are the **fold**,  $f(x) = \frac{1}{3}x^3 + ax$ , and the symmetrical **cusp**,  $f(x) = \frac{1}{4}x^4 + \frac{1}{2}ax^2$ .

# Geometries of the **fold** and **cusp** catastrophes.





For systems with a fixed potential we effectively have a **snap-shot** of parameter space. On average,  $r_f$  remains **close to unity** for many pathways in both model clusters and bulk, providing an explanation for **Hammond's postulate**.

**A Knotted Protein** (*PLoS Comput. Biol.*, **6**, e1000835, 2010)

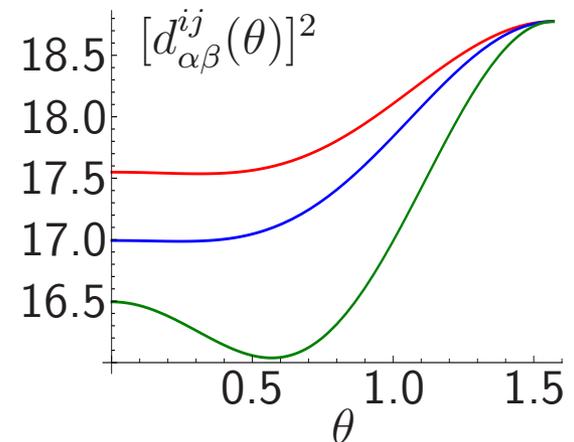
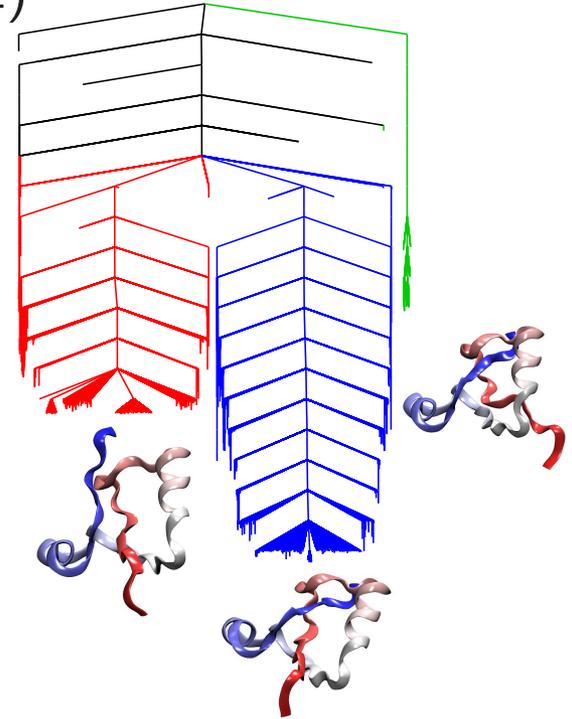
**Quasi-Continuous Interpolation** (*JCTC*, **8**, 5020, 2012)

The tRNA methyltransferase protein 1UAM contains a deep **trefoil knot** (right).

The folding pathway exhibits two **slipknot**-type steps for a truncated (residues 78–135) **Gō model** using an **associated memory Hamiltonian** and initial **QCI**.

The **QCI** potential **preserves** the covalent bonding framework, with short-range **repulsion** between **unconstrained** atoms. An **internal** minimum for **atoms**  $\alpha$  and  $\beta$  between **images**  $i$  and  $j$  occurs at

$$[d_{\alpha\beta}^{ij}(\theta)]^2 = \frac{|\mathbf{r}_{\alpha}^i - \mathbf{r}_{\beta}^i|^2 |\mathbf{r}_{\alpha}^j - \mathbf{r}_{\beta}^j|^2 - [(\mathbf{r}_{\alpha}^i - \mathbf{r}_{\beta}^i) \cdot (\mathbf{r}_{\alpha}^j - \mathbf{r}_{\beta}^j)]^2}{|\mathbf{r}_{\alpha}^i - \mathbf{r}_{\beta}^i - \mathbf{r}_{\alpha}^j + \mathbf{r}_{\beta}^j|^2}.$$



## Benchmarks for Landscape Exploration

**Minimisation:** Nocedal's algorithm, **LBFGS**, with line searches removed.

**Transition states:** single-ended searches use **hybrid eigenvector-following** (*PRB*, **59**, 3969, 1999; *JCP*, **111**, 7010, 1999; *CPL*, **341**, 185, 2001), double-ended searches use the **doubly-nudged** elastic band approach (*JCP*, **120**, 2082, 2004; **140**, 044115, 2014).

The **GMIN** (global optimisation), **OPTIM** (transition states and pathways) and **PATHSAMPLE** (discrete path sampling) programs are available from the **Cambridge Landscape Database** under the **Gnu** General Public License.

- Interfaces to many **electronic structure** codes are included.
- Current **svn** tarball image: <http://www-wales.ch.cam.ac.uk>
- <http://www-wales.ch.cam.ac.uk/tsbenchmarks.html> **Peptide** examples
- <http://theory.cm.utexas.edu/benchmarks/index.html> **OptBench** test suite
- <https://github.com/wales-group/examples> **Curated examples**